

Registration ID : 196804

PAPER ID : IJRAR19J1999

EDITOR IN CHIEF

R.B.Joshi

Volume 6 Issue 1 February-2019

Published In URMR (www.ijrar.org) ISSN UGC Approved & 5.75 Impact Factor

SBA-16 USING ULTRASONICATION

SYNTHESIS, CHARACTERIZATION OF RHA HIGH QUALITY NANO SIZED MESOPOROUS MATERIA

Shete S.B



International Journal of Research and Analytical Reviews (URAR)

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Synthesis, Characterization of RHA High Quality Nano Sized Mesoporous Material SBA-16 Using Ultrasonication

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Department of Physics, S.G.B. College, Purna (Jn) - 431 511.

Abstract: Spherical particles of mesoporous silica SBA-16 with cubic Im3m structure were synthesized at low pH using Pluronic F127 as template and TEOS as silica source. The diameter of the spherical particles can be controlled in the range of 0.5-8 µm by varying {Spherical particles of mesoporous silica SBA-16 with cubic Im3m structure were synthesized at low pH using Pluronic F127 as template and RHA as silica source. The diameter of the spherical particles can be controlled in the range of 0.5-8 µm by varying synthesis temperature t 45 °C. A sharp transition from large particle is observed. It is suggested that this morphology transition is due to a change in hydrolysis and condensation rate of the silica source and as a result the assembly of F127 micelles will differ. The SBA-16 samples were characterized using powder X-ray diffraction

(XRD), scanning electron microscopy (SEM), transmission electron microscopy (TEM) and Nitrogen adsorption

techniques.

Keywords: SBA-16; Spherical particles; Synthesis temperature; Morphology; F127

INTRODUCTION:

The synthesis of mesoporous materials by a hydrothermal treatment [1, 2, 5]. The properties of these materials make them attractive for adsorption, catalysis, separation, chemical sensing, optical coating, drug delivery and electronic applications. For practical purposes, the overall morphology of a mesoporous material is a necessary requirement in combination with their internal structure. For instance, in application such as high performance liquid chromatography isometric particles are required 20] and spherical particles are preferably used in chromatography for column packing as irregular particles tend to break down [22]. In this body-centered-cubic structure each mesoporous is connected with its eight nearest neighbors to form a multidirectional system of mesoporous network [23]. Due to its large cage, high surface area and high thermal stability, this material appears to be one of the best candidates for catalytic support and packing materials for separation. Using F127 as a surfactant is the common way of synthesizing SBA-16 [16, 24]. However, there are also reports on alternative surfactants such as F108 [25], a blend of P123 and F127 [26], and other nonionic surfactants [27]. Several studies have been carried out to understand the formation mechanism of this material, for instance, in the framework of the colloidal phase separation mechanism (CPSM) Yu et al. [29] suggested that, the formation process of mesoporous materials involves three stages: (1) ooperative self-assembly of inorganic/organic composites, (2) Spherical particles of mesoporous silica SBA-16 structure were synthesized at low pH using Pluronic F127 as emplate and TEOS

silica source Similarly, M. Mesa et al. [30] followed the formation mechanism of SBA-16 rom the earliest stage of the reaction until the formation of the particles by a dynamic light cattering technique, and suggested three steps for that: (1) silica coating of the surfactant icelles and crease of the zeta potential, (2) formation of micron-sized "liquid particles" by ggregation and fusion of the composite colloids (silica coated micelles), and (3) solidification of the "liquid particles" and transformation into the final particles of mesoporous silica. Synthesis conditions such as temperature, acidity, silica concentration, .This material can be prepared using varioussynthetic procedures, which can drastically influence itsfinal textural and morphological properties. If a ternary triblockcopolymer/water/butanol system is used, and the synthesis iscarried out at a lower acid concentration than in the raditionalcopolymer/water/HCl system, it is possible to enhance thestructural order over the final product. The synthesis is drivenby a self-assembling of the polymer in micelles[30], hich minimizesthe contact of its hydrophobic parts with the water solutionand leads to the information of the silica inorganic structure byhydrolysis of tetraethoxysilane (TEOS) at the ater/polymer

Experimental details:

Material synthesis: The synthesis of silica SBA-16 was made following a ternary system based on a previously described method reported inliterature.17 A solution of 2.5 g of Pluronic F127 (Sigma Aldrich) in a mixture of 120 g of distilled water, 5 g of concentrated HCl (37%) and 7.5 g of butanol was first prepared. After one hour stirring at 45 °C, 12 g of RHA were added and the mixture stirred for further 24 h. The ultrasonic treatment is given at 45 min. The step was followed by a hydrothermal treatment of 24 h at 100°C in Teflon coated stainless steel autoclave. Drying of the as synthesized sample was performed at room temperature overnight part of the dry sample was calcined under air flow by increasing the temperature 1.5°C per minute un to 550°C and maintaining it at this temperature for 6 h sample hereafter named calc-SBA-16 by Fabio

et al.,[1]

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Experimental:

XRD-studies:

The X-ray pattern of the As-Synthesized and synthesized mesoporous silica material is an highly periodic silica phases which is normally reflected by the distinct XRD signatures at low 20 angles from 1° to 10° as shown in Fig1 Sharp signal in XRD spectra indicates the presence of long range order of uniform hexagonal phase in the mesoporous materials. The well defined reflections from [110] plane are a prime characteristics of the hexagonal lattice symmetry of the SBA-16 structure with d_{10} value=26.3, 2θ =1.121, Intensity: 40000 cps I/I₀=100 Unit cell parameter a_0 = 37.8A⁰



Fig.1 XRD pattern of optimized RHA-SBA-16As-Synthesized and Synthesized Material

However, it is evident that RHA-SBA-16 samples synthesized using surfactant Pluronic F-127 in a range from 1g to 4g showed the corresponding peaks for the planes (110), (120) and (200) located at 1.5, 3.2 and 3.9° of 2θ respectively (Fig.1 Synthesized RHA-SBA-16).

Both, decreasing or increasing of surfactant Pluronic F-127 out of this range resulted in a decrease of the poremesostructured ordering due to the disturbing of the assembly on RHA-SBA-16 structure during process of synthesis. On the other hand, as the amount of surfactant Pluronic F-127 used was 2.5g all the three peaks are shifted to lower 20 values. This behaviour could be attributed to a thicker pore wall. In the case RHA-SBA-16 sample synthesized with surfactant Pluronic F-127 used less than 1.0g, it did not show the signals for the planes (110) and (200). This Pluronic F-127 behaviour is due to the increase in the surfactant concentration causing prohibition of unit cell growth and decrease in polymerization of silica resulting in poor hexagonal ordered RHA-SBA-16silica. **SEM and TEM-analysis:**

Fig.2 (A, B) (As-Synthesized and Synthesized RHA-SBA-16) presents typical micro graphs of the as-synthesized and calcined RHA-SBA-16 samples. Formation of the RHA-SBA-16 phase at hydrothermal conditions under autogenic pressure is seen to result, in shorter time, in a solid consisting of small agglomerates Fig.2 (A) (Synthesized RHA-SBA-16). The RHA-SBA-16 samples exhibit agglomerates with sizes between 2.7 and 19.2A⁰. Synthesized RHA-SBA-16 Sample exhibited uniform spherical shaped particles, with very small variations in the morphology and size might be associated with the differences in the synthesis gel conditions and hence in nucleation and crystallization.



(A)

(B)

Fig.2 (A, B)SEM images of As-Synthesized and Synthesized Material(RHA-SBA-16)

TEM-apalysis:

Fig.2 (C) represents the TEM images of **PHA SEAME**, **TEM** images of the parent samples provided strong evidence of the supports of the characteristic hexagonal skicate structures shown on TEM, supports the observation made by low and the RDs1d 1983 Shi Garu Buddhiowami Mahavidyalaya

PRINCIPAL



(C) TEM images of As-Synthesized and Synthesized Material (RHA-SBA-16)

Sorption studies

The adsorption/desorption isotherms of N₂shown in Fig. 2 and the values of specific BET surface area, mean diameter of the mesopores, total volume of pores and wall thickness obtained for calcined RHA-SBA-16 are presented in Table 1 are well in agreement with XRD analysis. Curves in Fig.2 and data from Table 1 shows an increase in the adsorption capacity of N₂ with time for the RHA-SBA-16samples synthesized, indicating an improvement in the order of the mesoporous phase, already observed by the increase in the XRD peak intensity (Fig. 2) According to IUPAC classification, the isotherm of RHA-SBA-16 is of type IV which is the characteristics of mesoporous material. For this sample, there is a continuous, slow adsorption of nitrogen after the point of inflection indicating a broad pore size distribution. The hysteresis loop is found over a wide range of relative pressures, p/p₉ (0.28-1.00). The shape of the hysteresis loop confirms the formation of a hexagonal phase. Thus, the XRD results are confirmed by N2-adsorption-desorption data, which are similar to those reported for SBA-16.

We noted that the surface areas of calcined RHA-SBA-16samples are comparatively more than as synthesized RHA-SBA-16.The more surface area (779.70m²/g) is an indication of well dispersion of active sites. The standards for the optimization of the sample are summarized in Table 1 along with calculated unit cell parameter, average pore diameter and



wall thickness etc.

Fig.3. : N2 adsorption-desorption isotherms of optimized RHA-SBA-16

Table 1 : Data obtained from XRD and N	2-sorption studies for optimized sample

đ110	Unit cellpara meter	S.A. (m²/g)	Avera ge pore diame ter (Å)	Pore volume (ml/g)	Average wall thickness (Å)
26.93	38.78	779.70	29.16	0.568	9.62
	d110	d110 Unit cellpara meter 26.93 38.78	d110Unit cellpara meterS.A. (m²/g)26.9338.78779.70	Unit cellpara meterS.A. (m²/g)Avera ge pore diame ter (Å)26.9338.78779.7029.16	d110Unit cellpara meterS.A. (m²/g)Avera ge pore diame ter (Å)Pore volume (ml/g)26.9338.78779.7029.160.568

Effect of Ultrasonic Irradiation Power

The ultrasonic irradiation synthesis of mesoperous molecular sieve RHA-SBA-16 Pursa investigated by drawn and polymeter or discussion of the section with molar condition. To study the officer of irradiation prodection initially the gel was prepared as described earlier in the section with molar composition as follows

MOLAR COMPOSITION :

0.91SiO2:0.0035 (F-127) : 2.08BuOH: 0.91HCI: 117H2O.

The prepared gel was treated with different ultrasonic irradiationpower 50W, 100W, 50W, 200W before optical conditions of hydrothermal method .The pH of the gel in each attempt of hydrothermal was maintained at 6.9 and hydrothermal temperature maintained at 80°C for 18h.

XRD-STUDIES:

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Figure 4 shows the powder XRD pattern of four calcined samples which exhibit a typical peak pattern with a very strong (110) diffraction peak and other (120) and (200) diffraction weak peaks. Although sample RHA-SBA-16 (50 W, 100 W) shows strong (110) diffraction peak and two (120), (200) diffraction weak peaks, but the higher angle peaks are not resolved by broadening may be due to loss of order. Also the (110) reflection was of weak intensity with inferior d₁₁₀ value indicating the existence of a predominantly amorphous content of silica in the products. As the power of ultrasonic radiation increased from 150W to 200W, the diffraction peaks were significantly strong. The XRD pattern of sample RHA-SBA-16 (200 W) exhibits typical three peaks along with a small fourth peak with notable intensity. Thus, it indicates that high quality of RHA-SBA-16 was synthesized. The data collected from the XRD analysis was tabulated in Table 2



Fig. 4: XRD-pattern of RHA-SBA-16 synthesized at various ultrasonic power irradiation

SORPTION STUDIES:



(A) (B) Fig. 5 : (A) Nitrogen adsorption and desorption isotherm of RHA-SBA-16 samples synthesized at various ultrasonic power irradiation and (B) Pore size distribution

The nitrogen adsorption-desorption isotherms for the ensured warper are shown in Figure 5 (A) and concepting pore distribution Figure 5(B). The N₂ adsorption isothermost samples BHA-SBA-16 (50 W, 100 W) examples a busines are shown in Figure 5 (A) and concepting steep corresponding to N₂ condensation in primary mesonades indicating a matter low degree of structural order of a structural order of a

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smaller pore size value. As the power of ultrasonic irradiation increases, it was found that the steps width corresponding to N2 condensation in primary mesopores decrease gradually and hysteresis loop become smaller which indicates that for the higher ultrasonic power there was increase in the structure consistency demonstrated by the sharp PSD curve.

Table 2 :	Effect of	f ultrasound	irradiation	power	on t	he st	tructural	and	textural	properties	of	RHA-SBA-16	at
synthesis	temperatu	ire 80°C							0.040000460				

d110 (Å)	Unit cell paramete r a _b (Å)	Specifi c Surfac e area (m ² g ⁻¹)	Pore Size (Å)	Pore volum e (cc/g)	Wall thicknes s (Å)	% Crystallanit y
23.96	34.50	803.11	25.93	0.502	8.57	47
24.16	34.79	816.59	26.10	0.537	8.69	71
25.02	36.03	878.25	26.49	0.544	9.53	89
25.30	36.42	954.22	26.53	0.568	9.89	100
	dine (Å) 23.96 24.16 25.02 25.30	d110 (Å) Unit cell paramete r a0(Å) 23.96 34.50 24.16 34.79 25.02 36.03 25.30 36.42	d110 (Å) Unit cell paramete r a0(Å) Specifi c Surfac e area (m ² g ⁻¹) 23.96 34.50 803.11 24.16 34.79 816.59 25.02 36.03 878.25 25.30 36.42 954.22	d110 (Å)Unit cell paramete r a0 (Å)Specifi c Surfac e area (m2g-1)Pore Size (Å)23.9634.50803.1125.9324.1634.79816.5926.1025.0236.03878.2526.4925.3036.42954.2226.53	d110 (Å)Unit cell paramete r a0 (Å)Specifi c Surfac e area (m2g-1)Pore Size (Å)Pore volum e (cc/g)23.9634.50803.1125.930.50224.1634.79816.5926.100.53725.0236.03878.2526.490.54425.3036.42954.2226.530.568	d110 (Å)Unit cell paramete r a0 (Å)Specifi c Surfac e area (m^2g^-1)Pore Size (Å)Pore volum e (cc/g)Wall thickness s (Å)23.9634.50803.1125.930.5028.5724.1634.79816.5926.100.5378.6925.0236.03878.2526.490.5449.5325.3036.42954.2226.530.5689.89

From Table 2, it is comprehensible that as the power of ultrasonic irradiation increases from 50 W to 200 W there was shift of the most intense peak to higher d110 spacing value as well as small variation in the pore diameter which is due .3 to increase in the wall thickness of the pores of RHA-SBA-16 molecular sieves, it results into higher specific surface area and pore volume, thermal stability of the product which may be due to special acoustic cavitation phenomenon of ultrasound radiations i.e. the formation, growth, and implosive collapse of bubbles in a liquid medium. During the formation of the framework of RHA-SBA-16, the ultrasound helps to disperse the small silica oligomers more homogeneously in the mixture. The cavitation phenomenon formed due to passage of acoustic vibrations accompanied by a few extreme effects, such as a local increase of temperature to about 5000 K and a local high pressure near 100 MPa which were called as hot spots. These hot spots within the surfactant-silicate interface accelerate the silica polymerization, which was slow and rate-limiting under normal conditions. The rate of formation of hot spots increases with increase in the power of ultrasound irradiation. This twofold function of ultrasound radiation results in the particles of sonochemical product being bigger and more aggregated than those prepared conventionally. Thus, the fabrication of the mesostructure can be achieved more efficiently. It was also worth noting that ultrasound radiation did not destroy the micellar structure. The Fig.6 shows the effect of ultrasonic irradiation power given to reaction gel, when treated with optimization condition of hydrothermal synthesis on crystallization of RHA-SBA-16 .It was observed with increase in irradiation power and reaches to maximum value (consider as 100%) as shown in figure.



Fig. 6 : Effect of Ultrasonic irradiation power on crystallization

SEM AND TEM ANALYSIS:

(150 W, 200 W) samples. Formation of Fig.2.31 (A) presents typical micrographs of the calcined RHA-SBA-16 the RHA-SBA-16 phase at hydrothermal cum ultrasonic conditions under autogenic pressure is seen to result, in shorter time, in solid consisting of small agglomerates Fig. 7 (A-RHA-SBA-16 (150 W, 200 W)).

The RHA 3BA-16 samples exhibit agglomerates with sizes between 24 and 270 nm. Sample RHA-SBA-16(150 W) exhibit agglomerates with very available and 270 nm. Sample RHA-SBA-16(150 W) with the relational spherical shaped particles, with very small variations in the morphology and size might be asso Shri Guru Buddhishani - 4315He unpurities in the extracted care from rice bask and hence in nucleation and crystallization. and orm spherical shaped particles, with very strict cariations in the morphology and size might be associated



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TEM-ANALYSIS:

(a)

Fig. 8 (shows spectacular the TEM image of calcined RHA-SBA-16. The mesopores are arranged in a honeycomb like structure, separated by thick, amorphous silica pore walls. From the TEM image, it was apparent that RHA-SBA-16 has a very large void fraction, due to the presence of mesopores.



Fig. 8: TEM images of Synthesized Material (RHA-SBA-16(200W))

FTIR-STUDIES:

The synthesized samples were further characterized by Fourier transform spectroscopy. The FTIR spectra of RHA-SBA-16 synthesized at various ultrasonic power irradiations illustrated in Fig. 9 The broad band around 3500 cm⁻¹ is due to surface silanol and adsorbed water molecules which indicating the silica framework is hydrophilic. The two peaks at 2922.9 cm⁻¹ and 2852.5 cm⁻¹ indicates the presence of organic template in the framework. RHA-SBA-16 tends to adsorb water vapours in air since the surface of silica framework is water liking; the stretching mode of H₂O is detected at 1634 cm⁻¹. Bands observed at 1234 cm⁻¹ and 1071 cm⁻¹ are characteristics peaks of asymmetric Si-O-Si stretching. Another characteristics peak is the symmetric Si-O-Si stretching recorded at 786 cm⁻¹. However, the peak at 2319 cm⁻¹ is prominently found to be changed due to the effect of ultrasonic power on the synthesis of RHA-SBA-16. The effect observed for RHA-SBA-16 is manifested by the XRD analysis and sorption studies of the samples.



Fig. 9 : FTIR spectras of RHA-SBA-16 samples synthesised at various ultrasonic irradiadition power

Effect of Utrasonic Irradiation time applied to reaction gel:

The initial gel was prepared as described earlier. The objected gel table 6.9) was stirred for 30 min at synthesis temperaturing of ollowed by ultrasound irradiation by Sonica of Level 256 by keeping the power constant i.e. 200W which work of imiged in proceeding section. Four samples were synthesized under different ultrasonic irradiation time of 5, 10 also and with the altrasonic power kept constant and the gel statistic for different time interval was subjected to purne the earlier section and were designated as RHA-SBA-16(Smit 10 min 15mit 2000) is not solved as the same as RHA-SBA-16(Smit 10 min 15mit 2000) is not solved as the same as manufacted in alays.

procedure followed is same as described earlier. The materials synthesized have been evaluated by x-ray diffraction and nitrogen sorption studies.

XRD-STUDIES:

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Figure 10 shows the XRD pattern of all the four samples synthesized for different ultrasound irradiation time interval and the corresponding data tabulated in Table 3. As shown in the Figure 10, the XRD pattern of sample RHA-SBA-16(5min)which was synthesized at irradiation time 5 min, the only reflection at (110) plane is more intense, it suggests that the material does not possess the well-defined hexagonal arrays even after the calcination. As the irradiation time is increased from 5 min to 15 min, the intensity of characteristic peak increased and the reflection (110) was shifted to lower 20 value consequently shifting of d₁₁₀ to higher value. The XRD pattern of calcined RHA-SBA-16 (20min) exhibit a typical four peak pattern with a very strong (110) reflection at low angle and other weaker reflections. The presence of (210) reflection suggest the excellent quality of synthesized RHA-SBA-16.If the irradiation time increased further there is gradual decrease in the value of d₁₁₀ spacing. Surprisingly dramatic decrease in the d₁₁₀ spacing decrease was observed for the irradiation time 30 min. This is may be due to the long-time ultrasonic irradiation minimizes the micellar size so that more but smaller micelles exist in the solution, resulting in an arrangement of silicate around the micelles and forming the mesophase structure with smaller unit cell size.



Fig. 10 : XRD pattern of calcined samples synthesised at various

ultrasound irradiation time

Sample designation	d110 (Å)	Unit cell paramete r a ₀ (Å)	Specifi c Surfac e area (m ² g ⁻¹)	Pore Size (Å)	Pore volum e (cc/g)	Wall thicknes s (Å)	% crystallanit y
RHA-SBA- 16(5min)	24.58	35.39	879.37	26.25	0.498	9.15	51
RHA-SBA- 16(10min)	25.00	36.00	893.22	26.74	0.513	9.26	73
RHA-SBA- 16(15min)	25.19	36.27	981.00	26.90	0.533	9.37	91
RHA-SBA- 16(20min)	25.30	36.42	954.22	26.53	0.568	9.89	100

Table 3: Effect of ultrasound irradiation time on the reaction gel

From Table 3, due to cavitation phenomenon of acoustic vibration of ultrasound waves in the gel, the specific surface area as well as pore wall thickness increases gradually as the irradiation time increases from 5 min to 20 min. Ultrasonic irradiation for longer time minimizes the micellar size so that more number of micelles exists in the solution but is of smaller size. Table 3 also summarizes the values of % crystallinity derived from X-ray diffraction pattern for various synthesizedRHA-SBA-16 samples which were further plotted as a function of ultrasonic power as shown in Figure2.34. The curve indicates the percent conversion from amorphous to 100 % crystalline product. Up to irradiation time 10 min the rate of conversion of amorphous to crystallized phase of RHA-SBA-16 was slow. Beyond that the rate increases rapidly due to formation of more hot spots within the surfactant-silicate interface which accelerate the silica polymerization in the synthesis gel. The degree of crystallization decreases as the process approaches to the completion indicated by constancy in percent crystallization. The apparent activation energy for conversion was estimated from the slope of linear plot of lnK vs (DFX by applying the Arrhenius equation d lnK/d (1/T) = Ew/RT, where K is the rate of conversion, Ew is the energy of stallization/ conversion, R is the universal gas constant. By the application of Arrhenius equation to the kinetics of brystallization of RHA-SBA-16, an apparent activation energy of conversion of aluminosilicate gel to 100% crystalline phase yas round to be 124.11kJmole" in the present crystallization of stent

SEM and Guru Bullu

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solid consisting of small agglomerates with sizes between 2.04 and 27.60 A⁰.Sample RHA-SBA-16 (20min) exhibited uniform spherical shaped particles, with very small variations in the morphology and size might be associated with the retention of some impurities in the extracted silica from rice husk and hence in nucleation and crystallization.



Fig.11 (A, B) SEM and TEM images of Synthesized Material RHA-SBA-16 (20min)

TEM-ANALYSIS:

Fig. 11 (B) shows brilliant TEM image of calcined RHA-SBA-16. The mesopores are arranged in a honeycomb like structure, separated by thick, amorphous silica pore walls. From the TEM image, it was apparent that RHA-SBA-16 has a very large void fraction, due to the presence of mesopores.

Conclusions:

Based upon the experimental study it was concluded that pure and ordered SBA16 material could be successfully synthesized by triblock copolymer F127 during 18 hrs of reaction. The parametric variations such as change of calcination temperature, the change of calcination time duration and the change of initial pH value of gel suggested that from RHA the well ordered mesoporous material SBA16 can be synthesized at 5500C for 4 hrs. keeping pH of gel 6.91. The maximum calculated surface area amounts to779.7 m2/g for the SBA16 materials keeping pH of gel 6.91, calcination time about 4 h. at 5500C.

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